



# **Laser Inertial Fusion Dry-Wall Materials Exposure to X-rays and Ions\***

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## Outline of Presentation

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- IFE Threat Spectra: RHEPP-1 and Z as simulators to ion/x-ray threat to first wall
- Brief Summary of Z exposure results - tungsten
- Description of RHEPP-1 and Heating/exposure Cycle
- Materials exposed on RHEPP-1 (ions):

### W and W alloys

Powder Metallurgy Tungsten (PM W)

Single Crystal W

Alloyed W - W25Re, W1%La

### Graphite/Carbon Fiber Composites (CFCs):

‘Engineered’ Materials as an alternative to flat wall

Carbon fiber ‘Velvet

W/Tac and W/HfC ‘Foams’

- Conclusions

# Summary of Exposure Results

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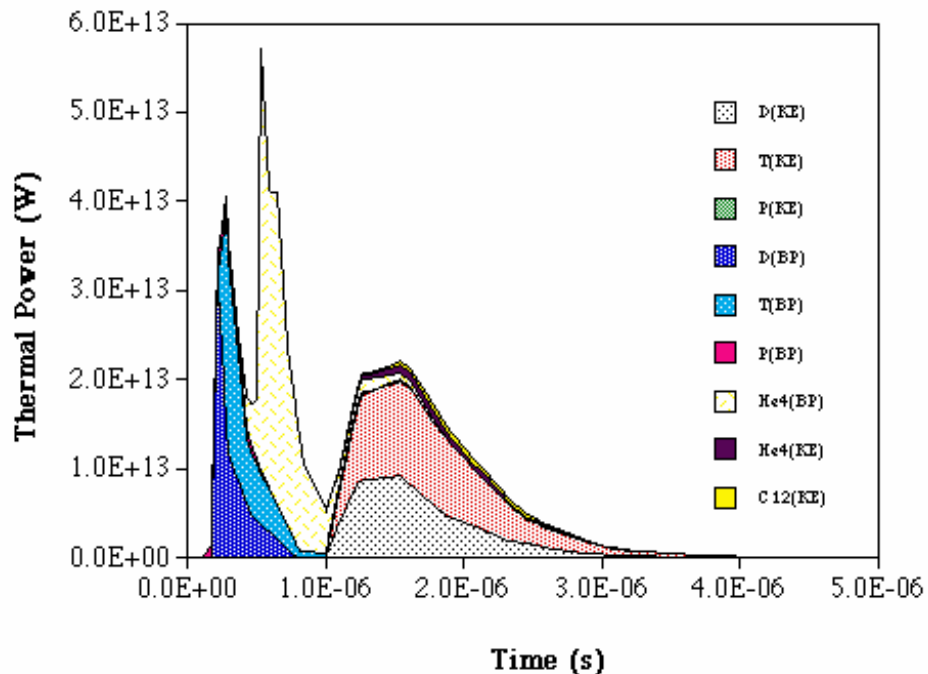
- Ion and X-ray exposure forms key part of Laser IFE Direct Drive First-Wall Threat. RHEPP-1 ion beams (multi-pulse) and Z x-rays (single-pulse) can simulate this threat with good fidelity
- Z results (X-rays on tungsten): melt threshold at  $1.3 \text{ J/cm}^2$ ; roughening threshold is grain-structure dependent, but  $< 1 \text{ J/cm}^2$

## RHEPP Results:

- For W materials, most serious issues are 1) Roughening below melt, evidently caused by thermomechanical stress, and 2) Stress cracking. Stress cracks deepen with pulse number, and may reach deep into interior. This is true regardless of melt. 'Divit' formation may indicate outright mass loss. Lower limit for PM W roughening is low  $\sim 1 \text{ J/cm}^2$ .
- For Graphite/Carbon Composites (CFCs), good survivability at low fluence, but rapid loss above sublimation. Physical sputtering or radiation-induced sublimation?
- Engineered materials such as 'Velvet' or 'Foam' are up-optimized so far, but may be an alternative surface design
- These results are also relevant to MFE W exposure due to Type I EL

## Laser IFE Direct Drive Threat Spectra

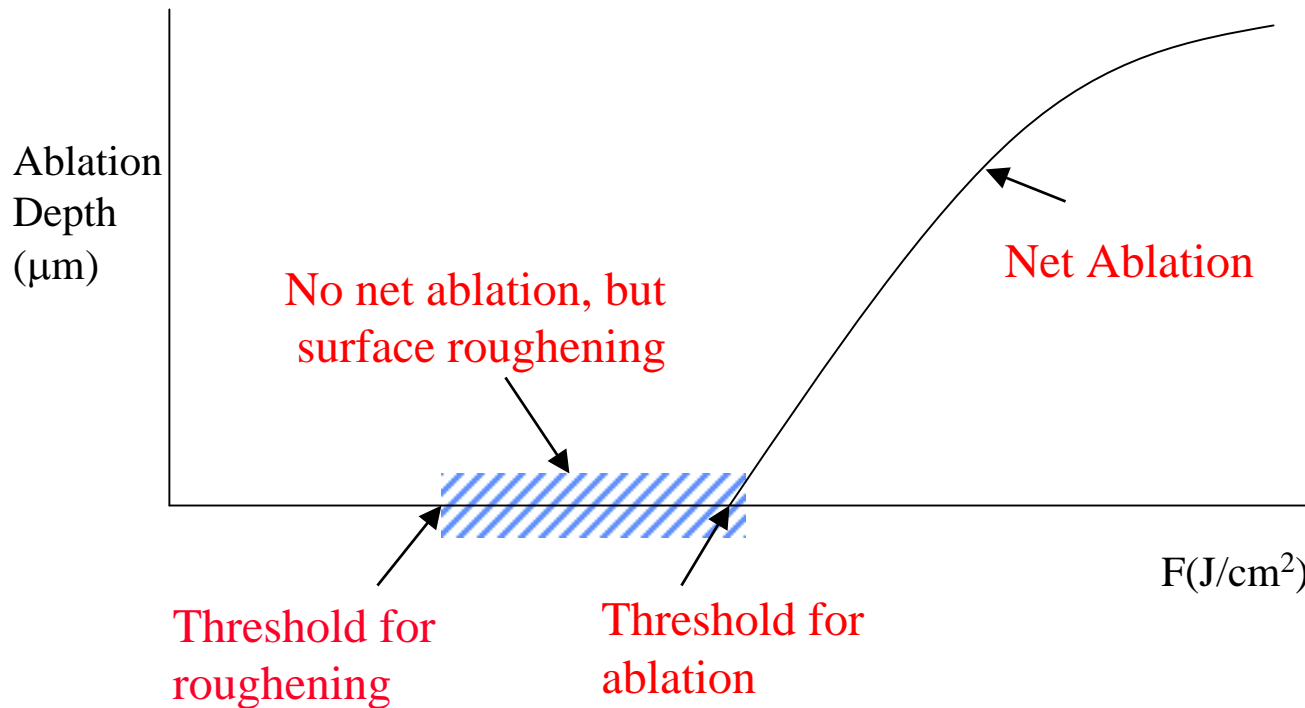
**Time-of-Flight Ion Power Spread**



**Simulation: Thermal Power to Wall in Ions  
from 154 MJ Yield. Wall Radius: 6.5 m**

- For Direct-drive Laser IFE:  
1-2% x-rays  
30% ions (50-50 fusion and 'debris')  
~70% neutrons
- Ions: several MeV, ~ 0.5  $\mu$ sec each,  
8-20 J/cm<sup>2</sup> fluence, judged  
Significant Threat
- X-rays: ~ 1 J/cm<sup>2</sup>, up to 10 keV energies,  
judged less significant threat
- Z: up to 10 keV x-rays, good single-shot  
simulator
- RHEPP-1: 800 keV He, higher for N<sup>2+</sup>,  
100-300 ns pulsewidth
- RHEPP-1 energy delivery too short, but  
otherwise good fidelity with reactor ion threat

# Regimes of IFE Materials Response Studies for x-rays and ions



Goals (for each material): examine net ablation to validate codes  
find threshold for ablation

**Area of Interest**

understand roughening  
find threshold for roughening

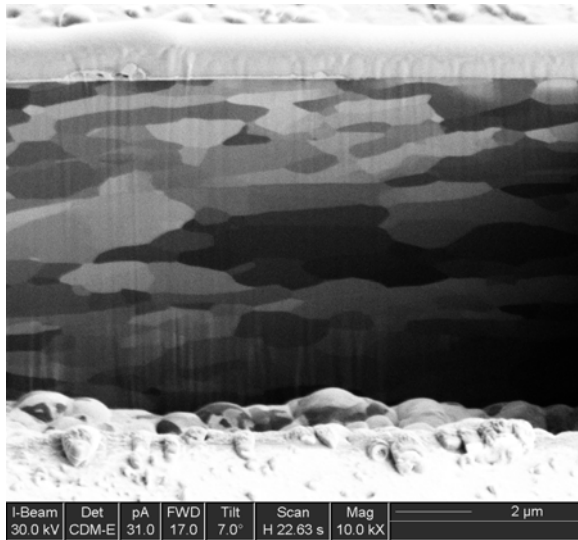
## Z Results - PM W Melting/Roughening Thresholds

### Tungsten Exposure on Z: Melting and Roughening Thresholds

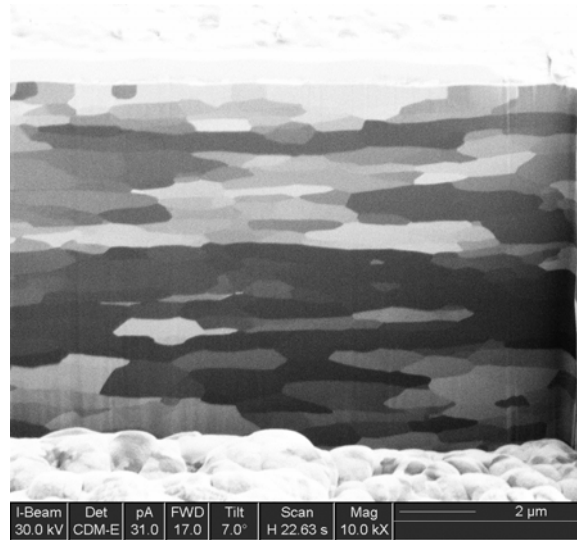
- Tungsten samples exposed to various fluence levels in Z machine with single shots.
  - Varied source-to-sample distance
  - Filter material: 8 $\mu$ m Be and 2 $\mu$ m Mylar
- Polished tungsten (Snead) prepared in 3 different ways:
  - single crystal
  - rolled powdered metal
  - chemical vapor deposition
- Latest samples preheated to 600° C
- Surfaces analyzed with
  - optical surface profilometer
  - Scanning electron microscope (with backscatter detector)
  - focused ion beam

**Z Results - PM W Melting Thresholds**

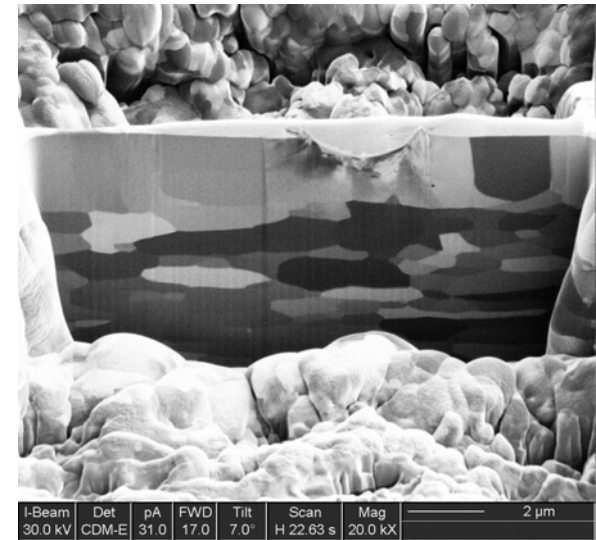
**Single-pulse Z data shows no melting below 1.3 J/cm<sup>2</sup>**



**1.3 J/cm<sup>2</sup>**  
**2 m kimfoil**  
**2.5 m Be**  
**0.1 m Al**  
**No melting**



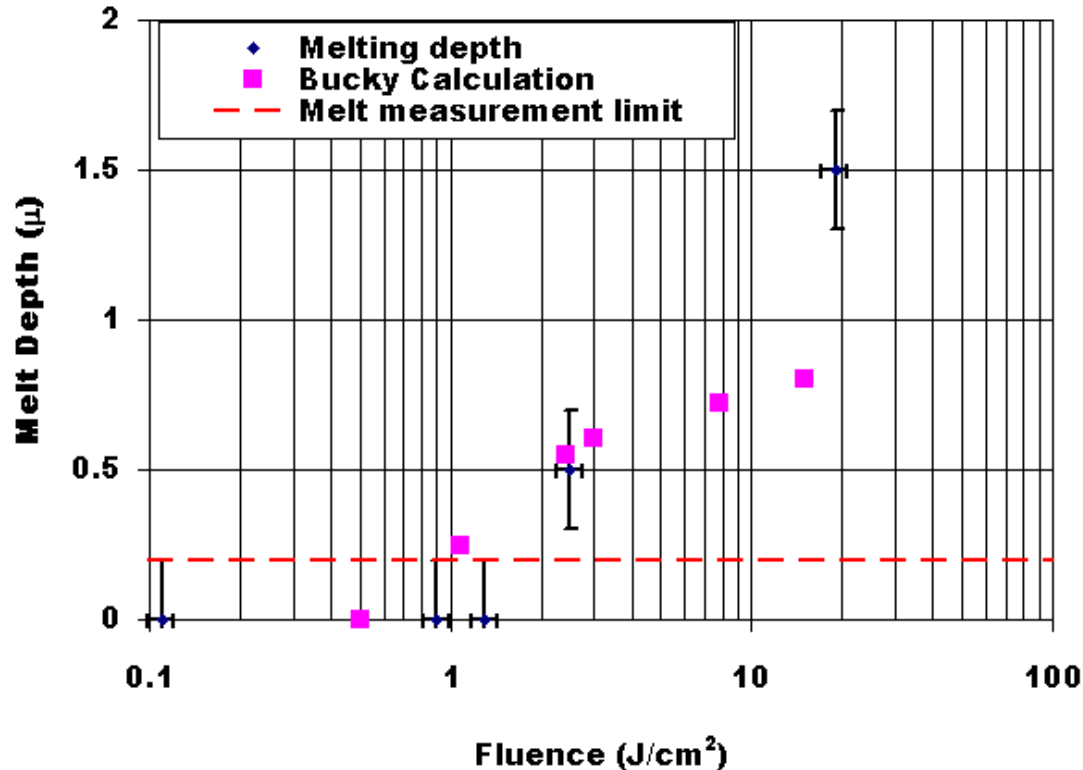
**2.3 J/cm<sup>2</sup>**  
**2 m kimfoil**  
**0.1 m Al**  
**0.5 μ melting**



**19 J/cm<sup>2</sup>**  
**2 μ melting**

- Unheated W
- Difficult to assess roughening with Z samples

## Melt depth vs. Fluence on Z

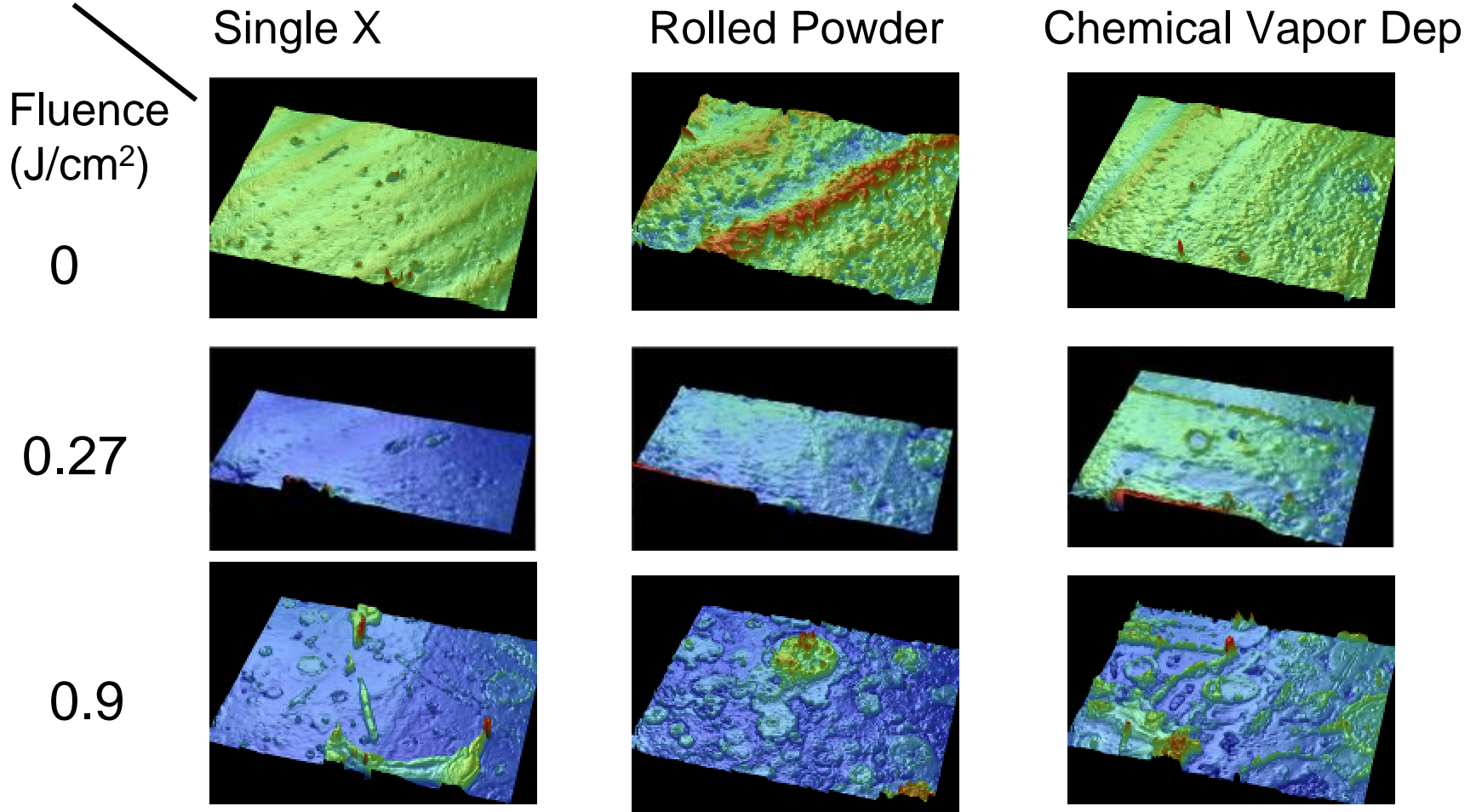


- Experimental points (error bars) and BUCKY calculation (squares) using filtered fluence numbers.
- For melt depths  $< 0.2 \mu$  (dashed line), melt depth  $<$  crystal grain size, so may or may not result in observable effect



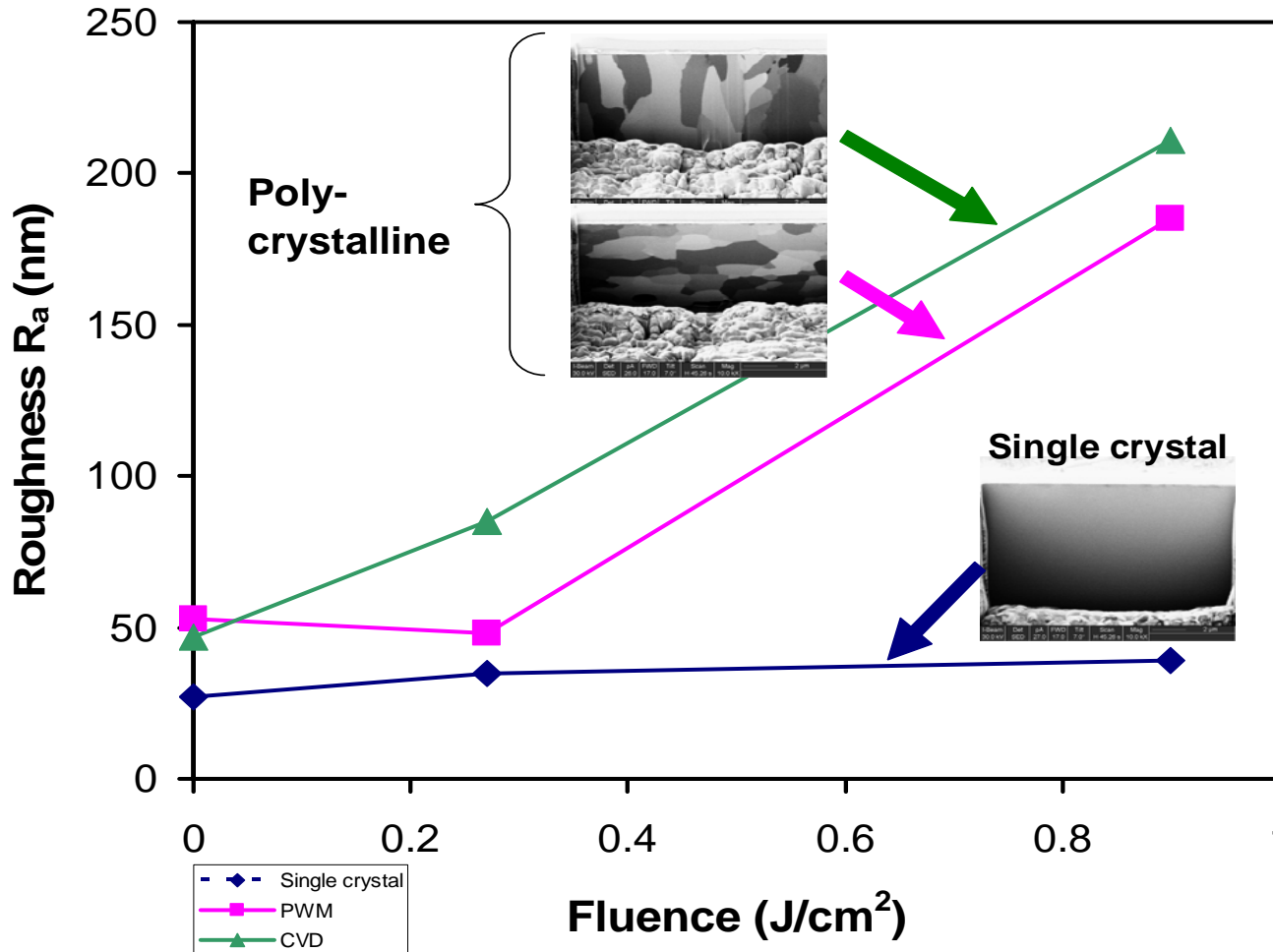
Z Results - PM W Roughening Thresholds

Images from samples on Z from VEECO surface profiler



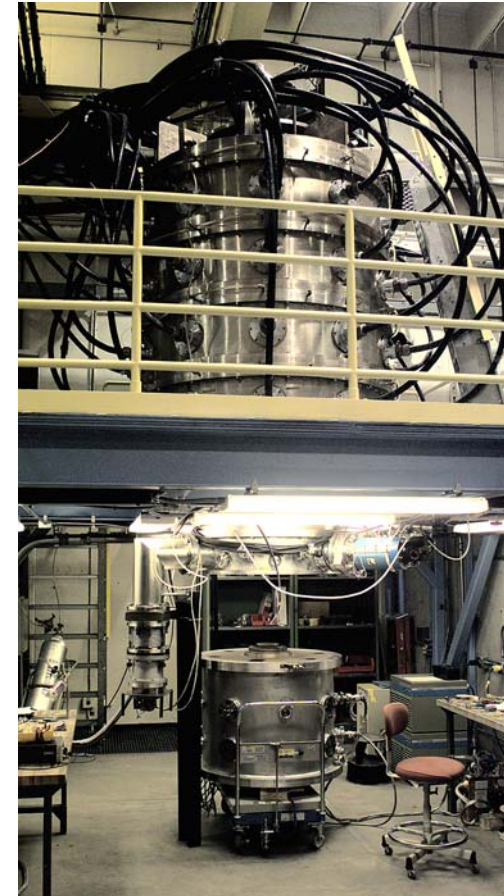
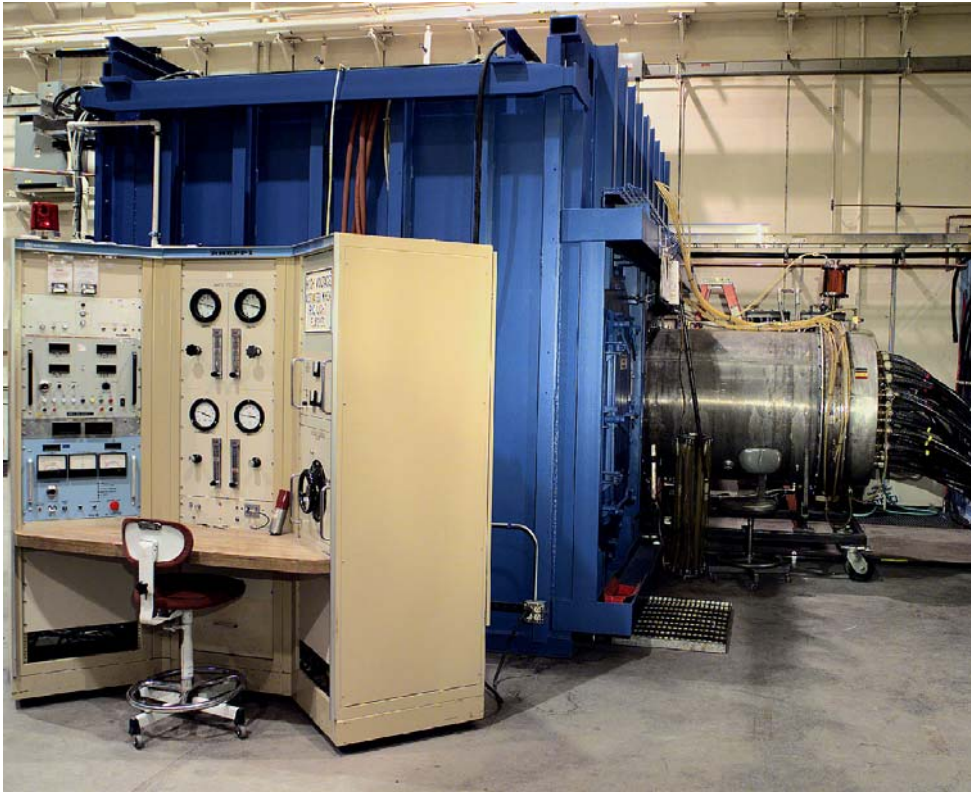
## Z Results - PM W Roughening Thresholds

### Roughening Depends on Fluence and Grain Boundaries



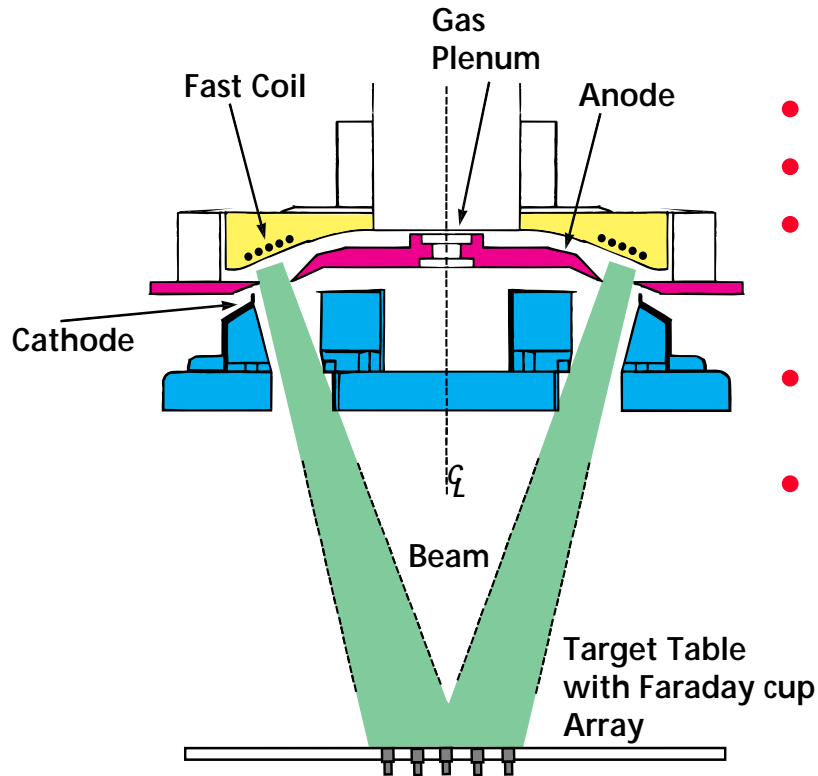
Roughening threshold:  $\sim 0.9 J/cm^2$  - single crystal, and  $\sim < 0.3 J/cm^2$  for polycrystalline tungsten

## RHEPP-1 combines Repetitive High Energy Pulse Power with a robust and versatile ion source: MAP



Left: Marx tank with pulse-forming line  
Right: 4-Stage LIVA and treatment chamber

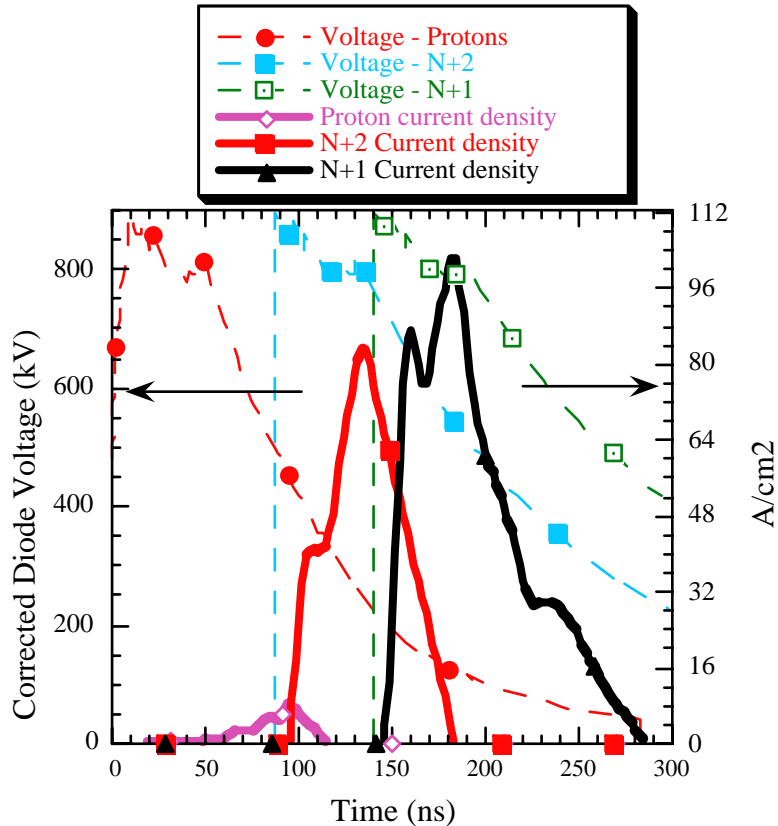
# The MAP (Magnetically Confined Anode Plasma) Ion Source is used for surface modification experiments on RHEPP-1



- 600-800 kV
- $< 250 \text{ A/cm}^2$
- Beams from H, He,  $\text{N}_2$ ,  $\text{O}_2$ , Ne, Ar, Xe, Kr,  $\text{CH}_4$
- Overall treatment area  $\sim 100 \text{ cm}^2$
- Diode vacuum  $\sim 10^{-5} \text{ Torr}$



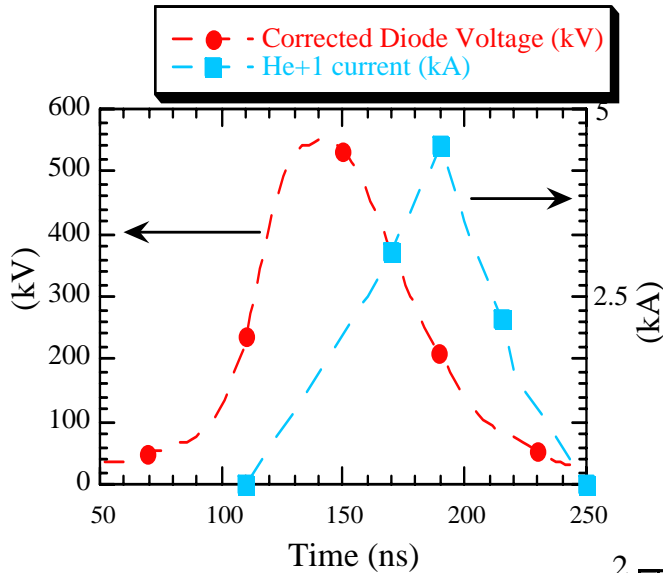
# Nitrogen injection into MAP produces 3-component beam of mostly N<sup>++</sup>, N<sup>+</sup>



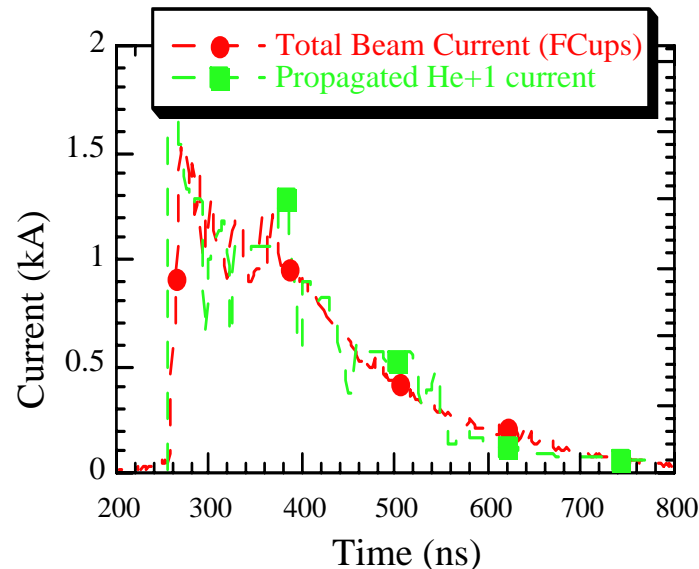
Shot 31661

- Beam predominantly N<sup>++</sup> and N<sup>+</sup> after small proton pulse at front
- Peak voltage = 850 kV  
Peak current density (total) ~145 A/cm<sup>2</sup>
- Total fluence = 7.9 J/cm<sup>2</sup> - will ablate almost all materials
- Total pulse width ~ 200 ns
- Ion range (TRIM):
  - N<sup>+</sup> 0.9 μm, N<sup>++</sup> 1.2 μm
- Oxygen, Neon beams similar

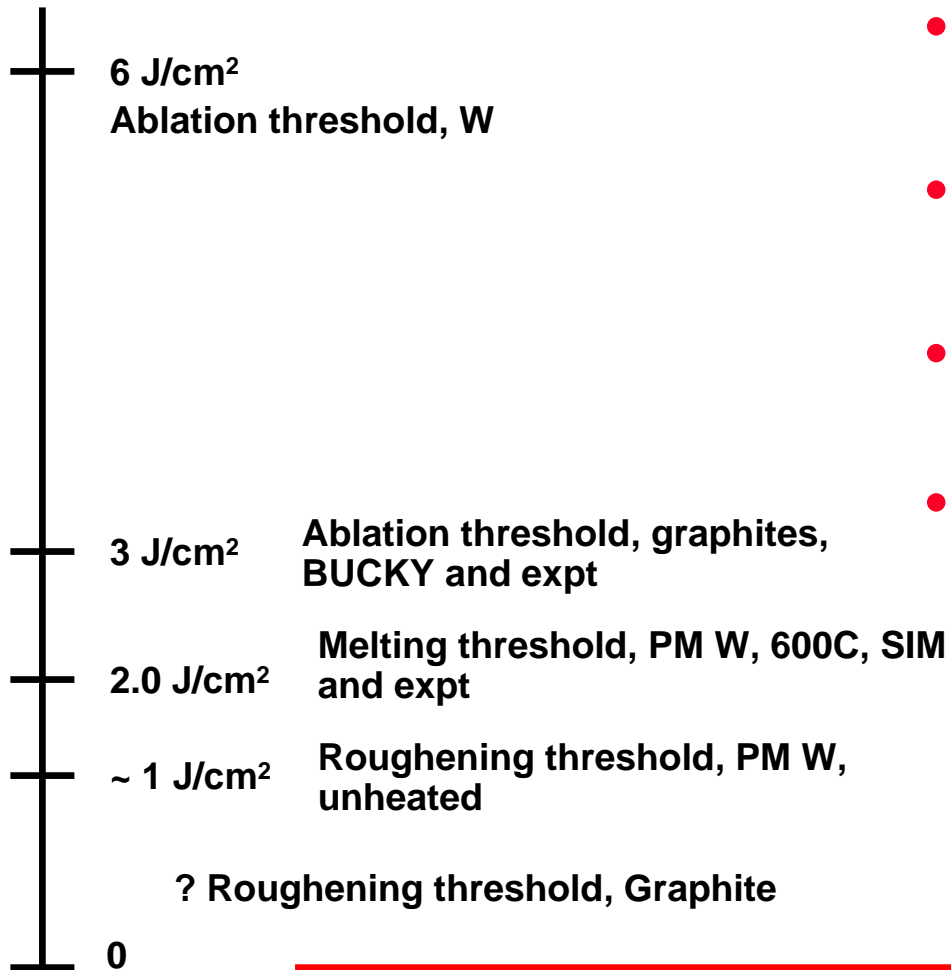
# He beam ( $\text{He}^{+1}$ ) pulsewidth is longer than N or C beam



- He current lags voltage, leading to debunching
- Current pulse width at 63 cm is almost 0.5  $\mu\text{sec}$  long
- Beam here was intentionally attenuated
- He range in W (TRIM)  $\sim 0.9 \mu\text{m}$
- Range similar to N beam because singly charged



## Thresholds for Materials exposure to ions on RHEPP

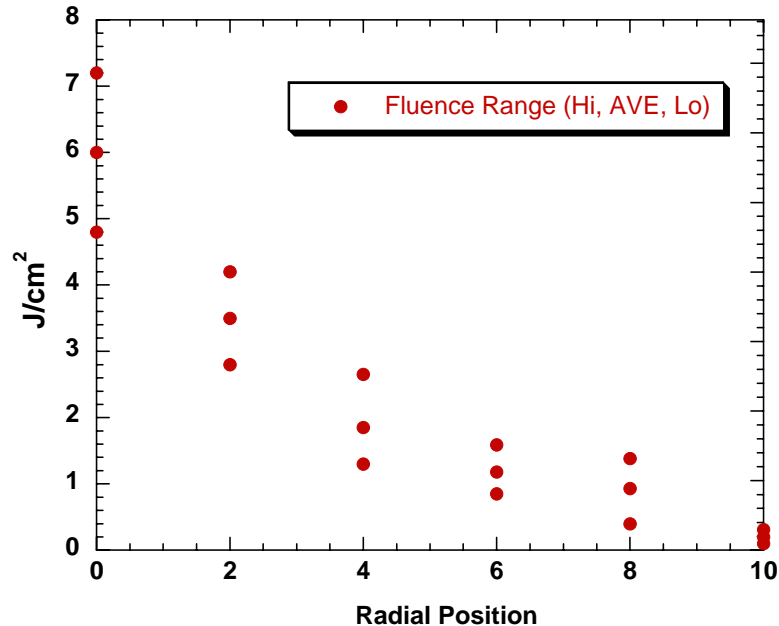


- General exposure conditions: MAP nitrogen beam, 150 ns pulsewidth, single shot.
- Roughening threshold for graphite (matrix) depends upon graphite
- Roughening threshold for W for He beam is below that for MAP N
- Roughening threshold, unheated, for  
W25Re: 3.5 J/cm<sup>2</sup>  
Re: 1 J/cm<sup>2</sup>

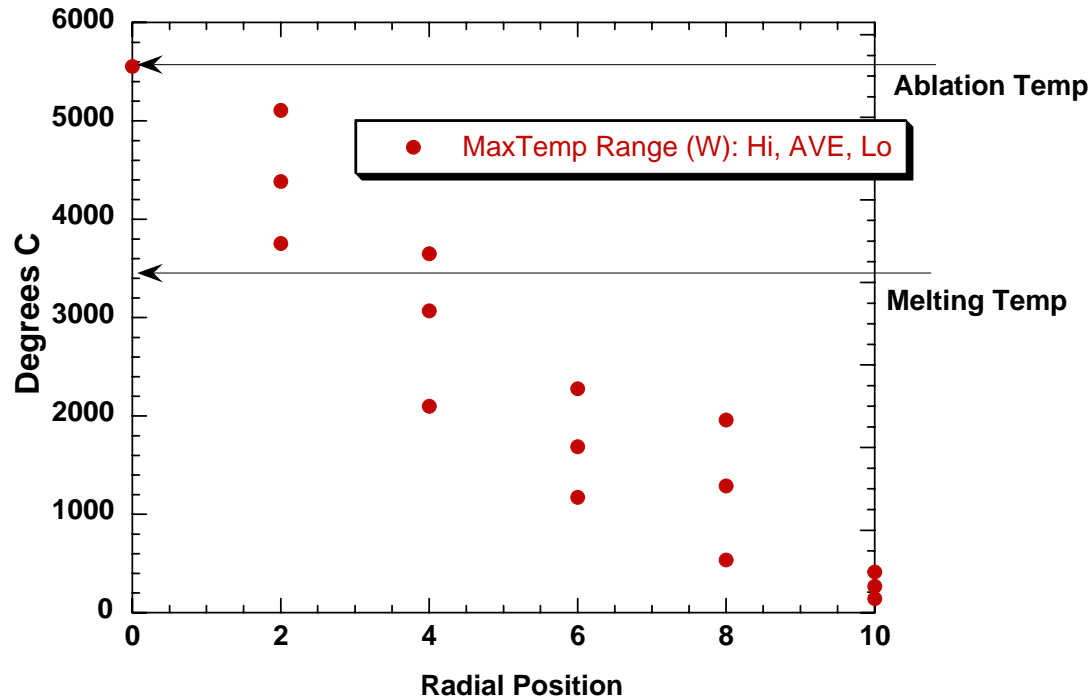
Based on AVERAGE fluences from RHEPP

# RHEPP-1 Surface Roughening

## Range of Fluences, MaxTemps (W) for RHEPP-1



Fcups averaged over 10 or 40 pulses

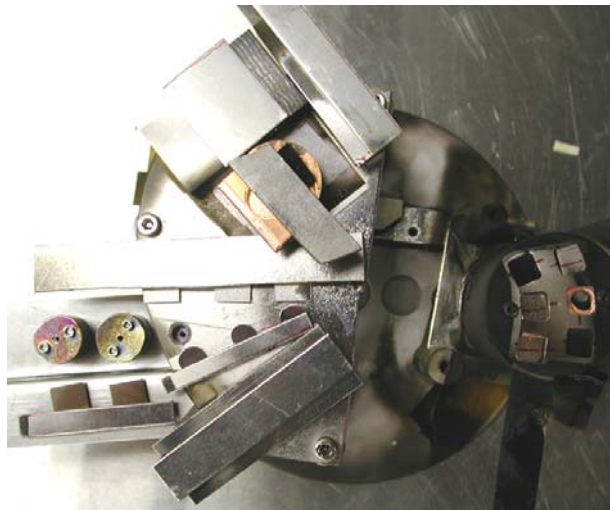


TEMP simulations (SIM CODE)  
Based on Fcups (Room Temp)



## RHEPP-1 Ion Exposure

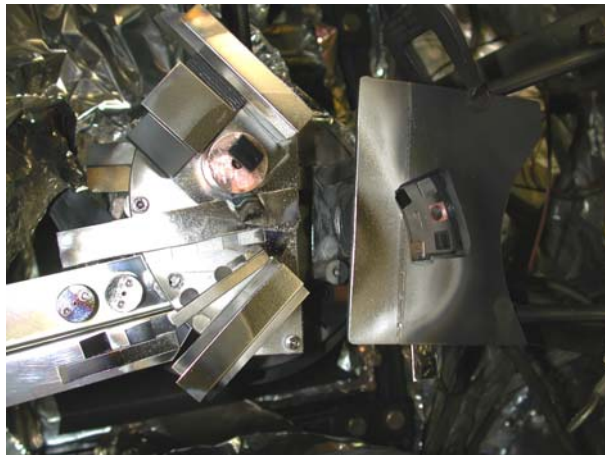
# Setup for current 2000 pulse Series: Metals and graphite, RT and 520C Roughening and fatigue propagation study



Before



Heated

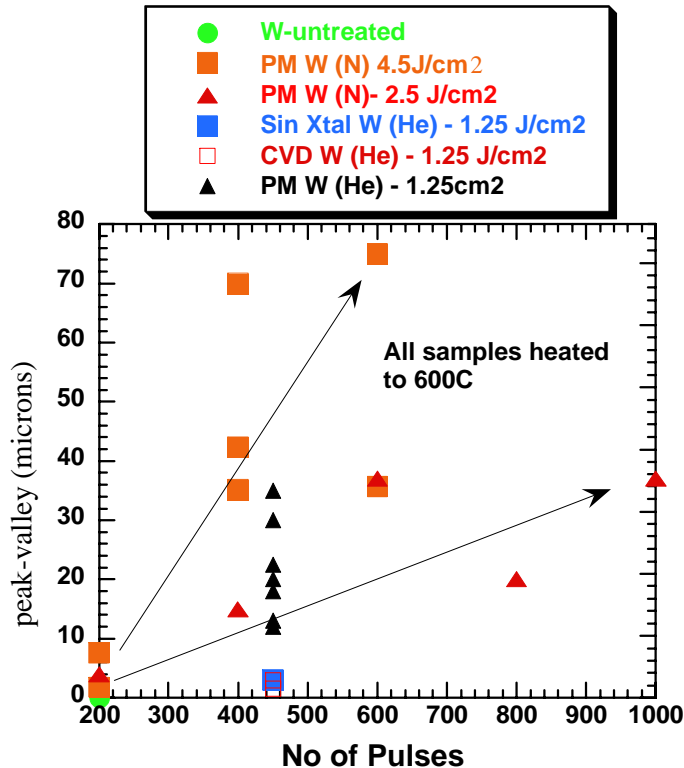


After 400 pulses

- **Goals:** Test below-melt response, explore if roughening saturates, study crack propagation, 2000 pulses total
- **Upper Photo:** 6-element heated samples (right), RT samples (left). Heated samples below melting, RT below/above
- **Heated samples:** CFC NB31, PM W (Snead), W25Re, graphite R6650, Sing XtalW(Snead), WC. C samples from FZ Julich (Linke). Velvet added later
- **Samples shot 400X, then SEM, Dektak (Renk/Tanaka), then reloaded for another 400 shots.**
- **RT samples:** PM W, Sing Xtal W, NB31, R6650, Ti-6Al-4V, W25Re, Re

## RHEPP-1 Surface Roughening

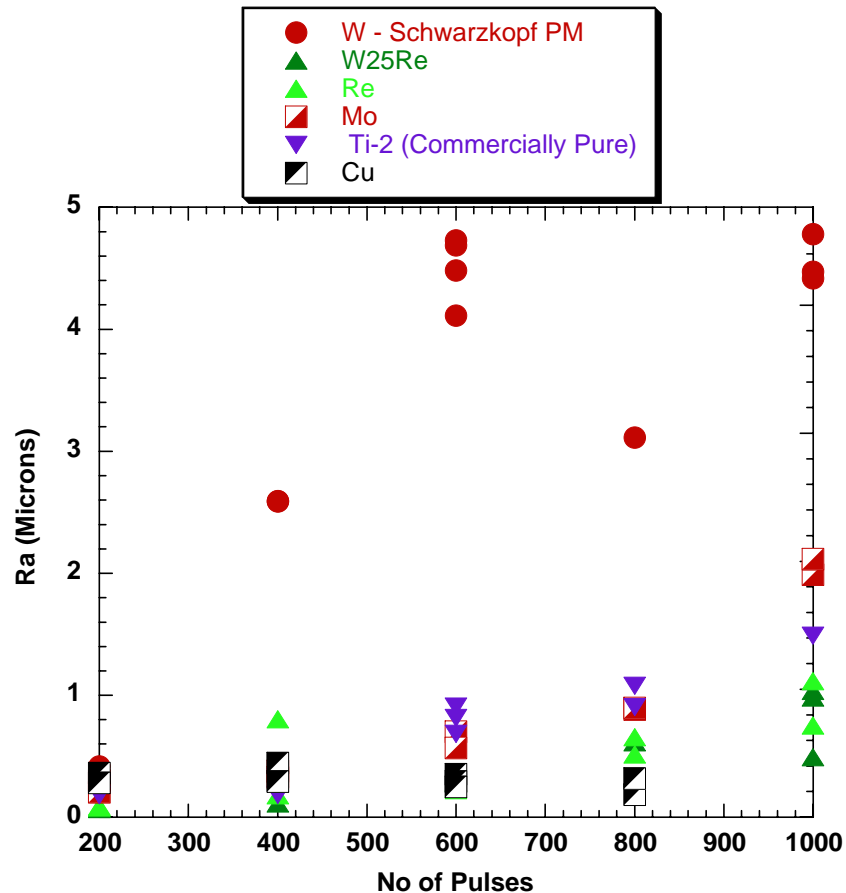
**Surface Roughening: Develops over many pulses  
Increases with fluence/shot number; PM W the worst**



- Roughening increases ~ linearly with pulse number
- Roughening increases with fluence per pulse
- Very little surface relief happens until after 200 pulses
- 4.5 J/cm<sup>2</sup> is between melt and ablation for W
- He beam roughens more than N (black points)
- CVD, SingXtal (actually, everything) roughens less than PM W
- W Peak-Valley exceeds 70 μm at 600 shots
- Is W roughening reaching saturation?

## RHEPP-1 Surface Roughening

Evolution of  $R_a$  Roughness at  $2.5 \text{ J/cm}^2$ , various metals :  
PM W is worst, then everything else. Cu does NOT roughen

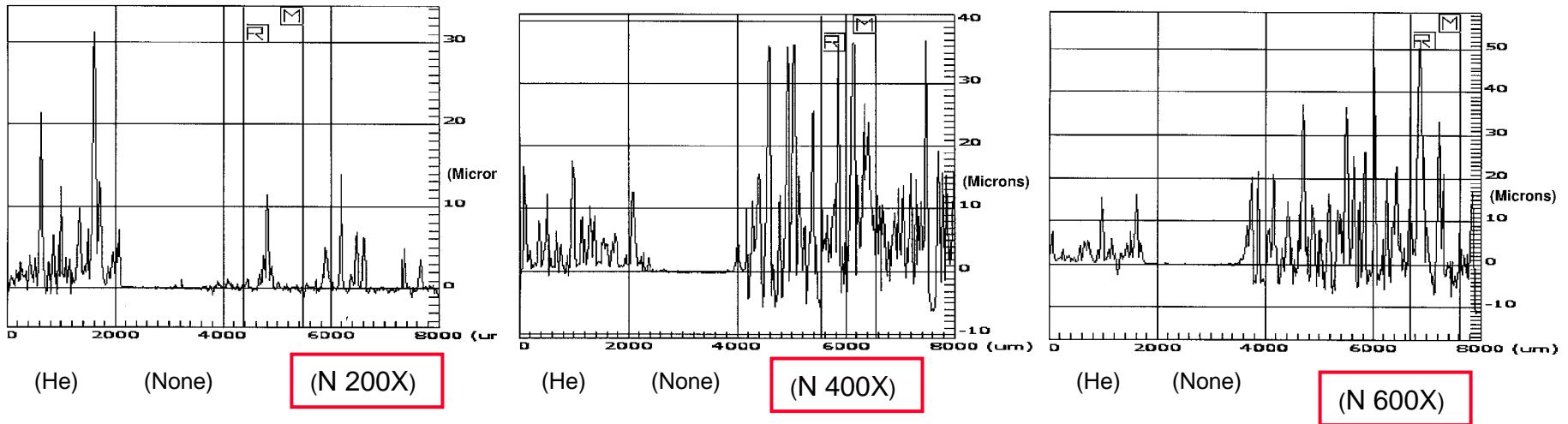


- $2.5 \text{ J/cm}^2$  is near or above melt for most metals shown
- Polished W and Mo are heated to 550-600C
- MO reaches  $2 \mu\text{m}$  Ra at 1000 shots, but Cu remains very flat
- Ti-2, Mo roughen steadily to 1000 shots
- W Peak-Valley exceeds  $35 \mu\text{m}$  at 600 shots, then plateaus. Is this a saturation point?

**RHEPP-1 PM W Surface Roughening**

**$R_a$  as function of Number of Nitrogen pulses @ 4.0 J/cm<sup>2</sup>:  
(Right) - Roughness increases from 200-400-600 pulses**

I-D 8 mm Profilometer Scans, 450x He (Left) and Nitrogen (Right)



Ra ~ 1-3 μm  
P-V ~ 10-30 μm

Ra ~ 1-3 μm  
P-V ~ 5-15 μm

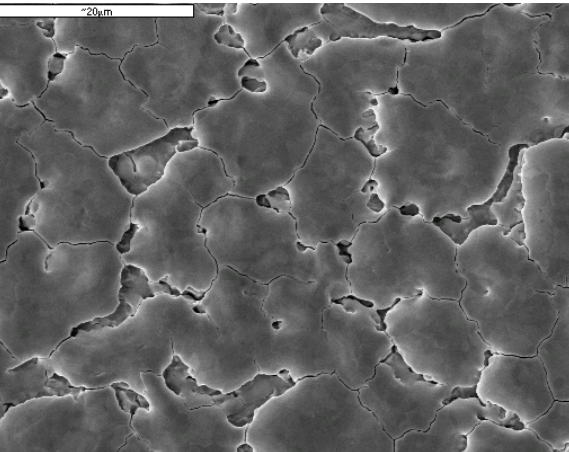
Ra ~ 4-9 μm  
P-V ~ 20-35 μm

Ra ~ 7-10 μm  
P-V ~ 50-70 μm

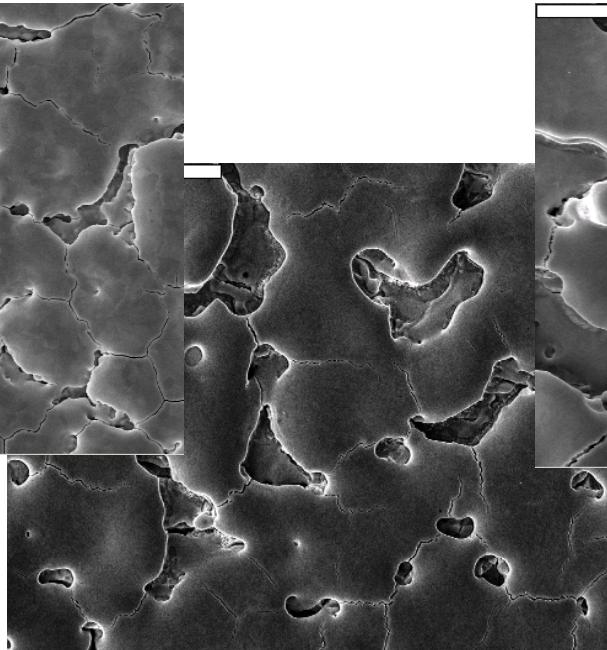
**Left Side (He) same in ALL plots. Material: Snead PM W**

## RHEPP-1 PM W Surface Roughening

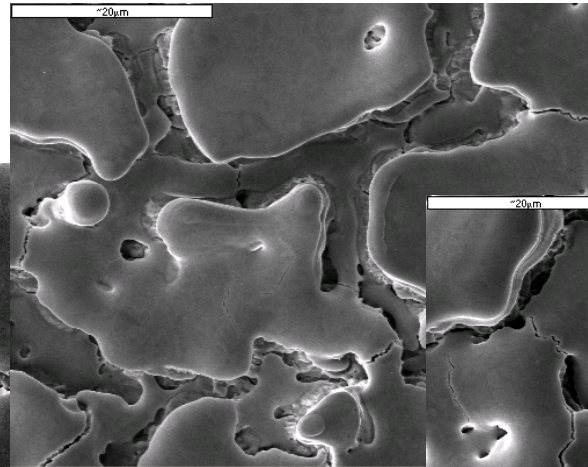
**SEMs of PM W (non-melt): appears stress cracking starts, then exfoliation, forming valleys**



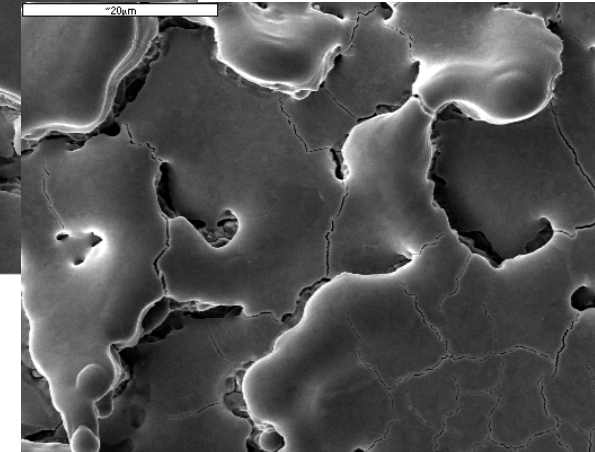
400 pulses



800 pulses



1200 pulses



1600 pulses

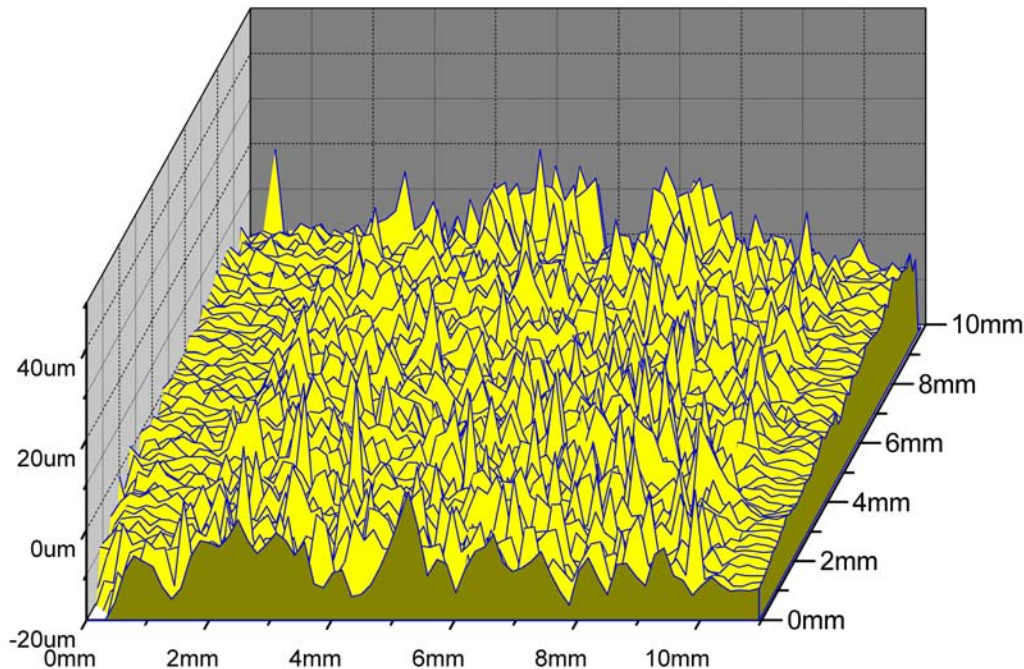
All images 2000X

- Heated PM W (600C) exposed to N beam at  $\sim 1.5 \text{ J/cm}^2$  - peak temp  $\sim 3300\text{K}$
- Rounded 'knobs' are actually high points. But does average surface height rise or fall?

## RHEPP-1 PM W Surface Roughening

# PM Tungsten after 1600 pulses (non-melting): Mostly mountains

Tungsten 1600 Pulses

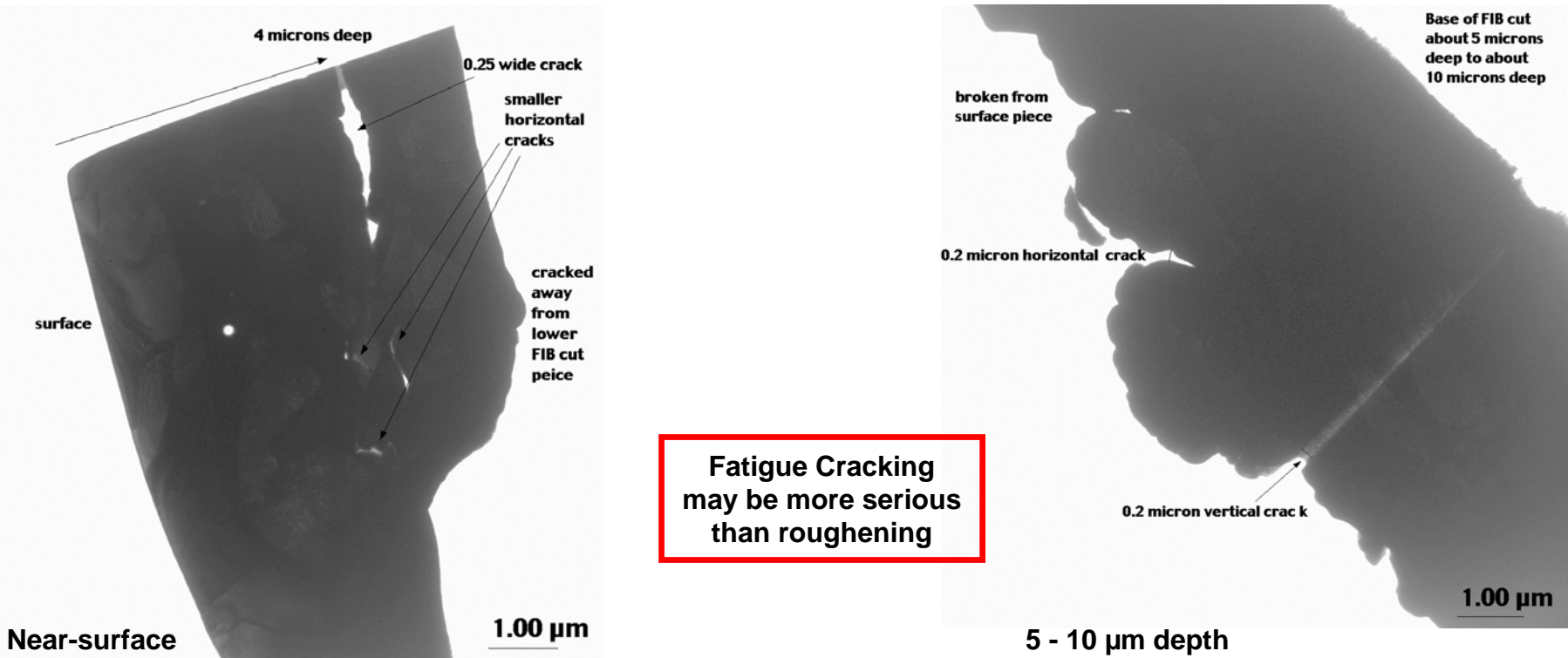


- Heated/treated PM W examined with NEXIV laser interferometry
- Comprehensive line-out scan: max height 30  $\mu\text{m}$ , min height < 10  $\mu\text{m}$  compared to untreated
- Very deep microcracking is present but not visible here

**Exposure Conditions:**  
Ave 1.45 J/cm<sup>2</sup>  
Hi 1.71, Lo 1.19  
MaxTemps: Ave 2900C  
Hi 3275C, Lo 2450C

## RHEPP-1 PM W Fatigue Cracking

### FIB-XTEM of 1000-pulse W at 2.25 J/cm<sup>2</sup> (ave): Deep horizontal/vertical cracking without melt

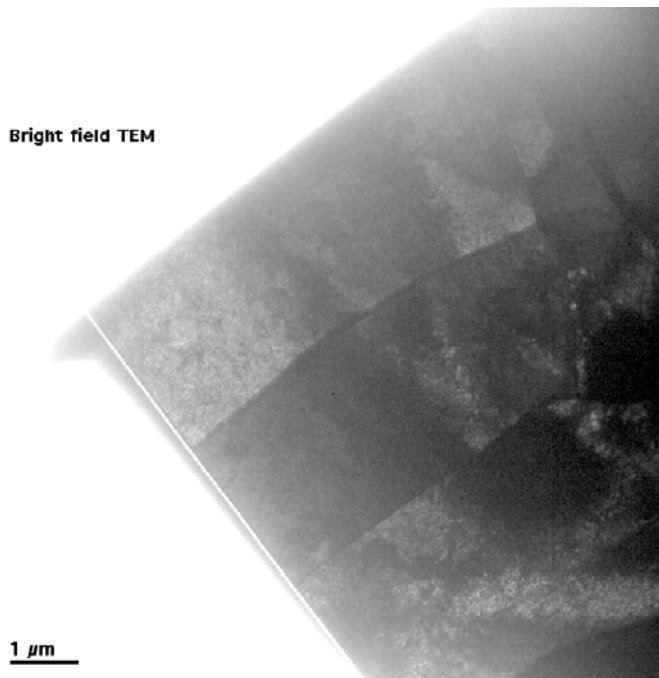


- Polished Powder Met W exposed to 100 shots N beam @ 2.25 J/cm<sup>2</sup> ave /pulse, ~ melting temperature at surface. No melt layer observed.
- 600°C exposure
- Sample cracking horizontally/vertically down to 10 μm depth
- Suspect fatigue-cracking

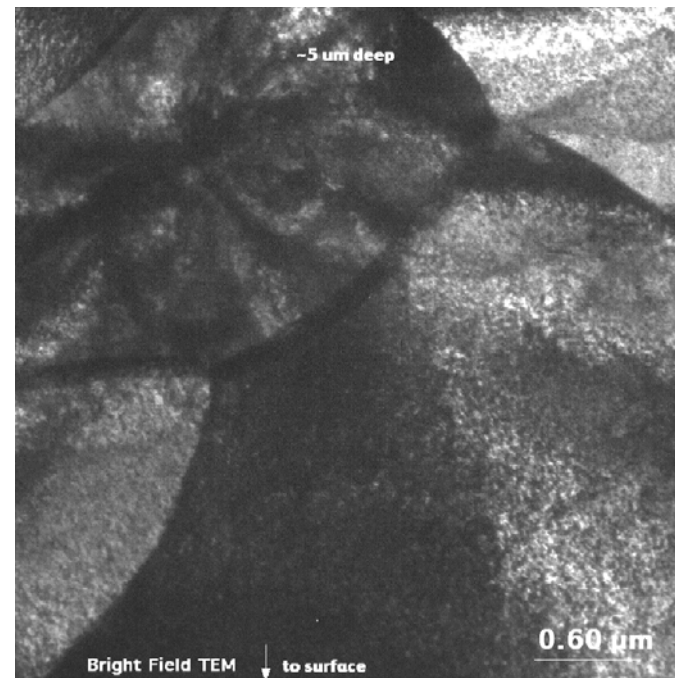
**RHEPP-1 PM W Fatigue Cracking**

**Fatigue Cracking only evident after extended exposure -  
XTEM of 60X-treated PM W (right) show no cracking in depth**

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Bright-Field TEM image



TEM image from 5  $\mu\text{m}$  depth

**60 pulses at ablation threshold:  
No cracks**



**RHEPP-1 Roughening Threshold, W, Multi-pulse**

**New 2000Series confirms multi-pulse roughening threshold for PM W at ~ 1 J/cm<sup>2</sup> (RT). R<sub>a</sub> increases rapidly with fluence**



PM W, N6 2272C\_AP (Collar) PM  
510C, 1.1 Hi 2830C W, 400  
J/cm2 Lo 1515C pulses  
800 pulses



PM W, S10  
0.2 J/cm<sup>2</sup>  
270C\_AP  
Hi 415C  
Lo 145C  
400 pulses

PM W, S8  
0.9 J/cm<sup>2</sup>  
1290C\_AP  
Hi 1960C  
Lo 535C  
400 pulses

PM W, S6  
1.2 J/cm<sup>2</sup>  
1690C\_AP  
Hi 2278C (5%)  
Lo 1175C  
(5%)  
800 pulses

PM W, S4  
1.9 J/cm<sup>2</sup>  
3070C\_AP  
Hi 3650C  
Lo 2100C  
800 pulses

Graphite  
R6650  
400/800  
pulses

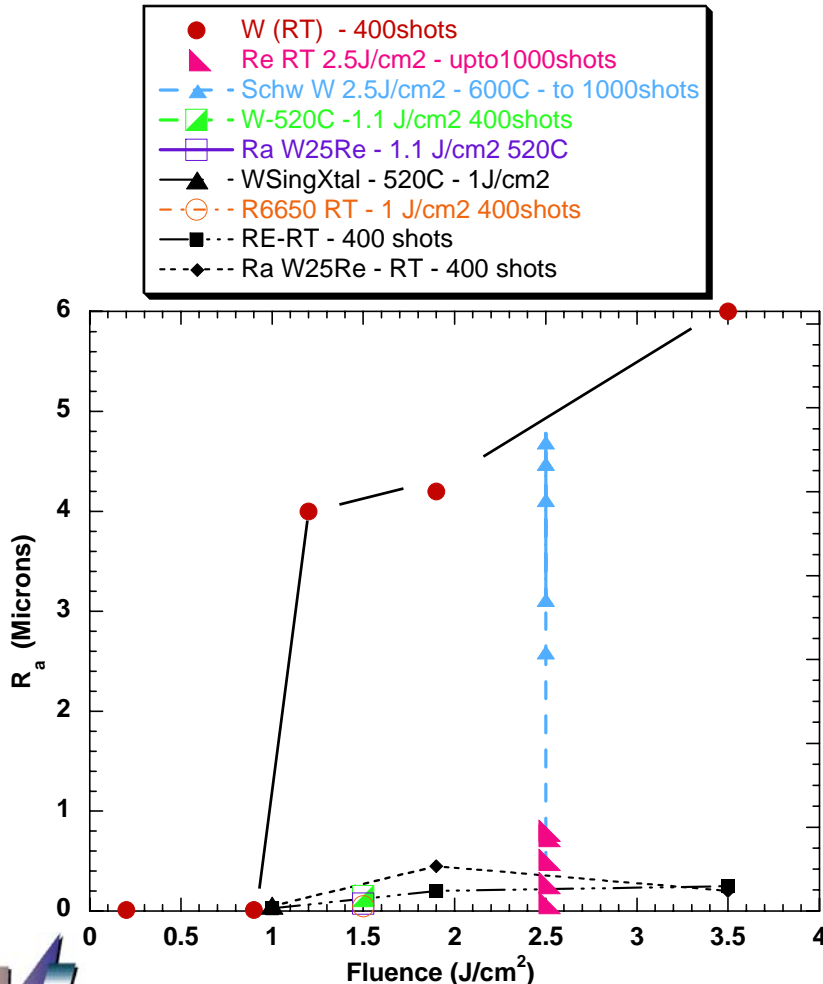
**These Snead PM W appear unaffected**

**These Snead PM W are very rough**

AP = Average High Surface Temp  
Hi and Lo occur on 5% of pulses

RHEPP-1 Surface Roughening, Current Series

Evolution of  $R_a$  Roughness , current series plus previous data:  
PM W RT roughening severe, heating helps, but trend w/shots is up

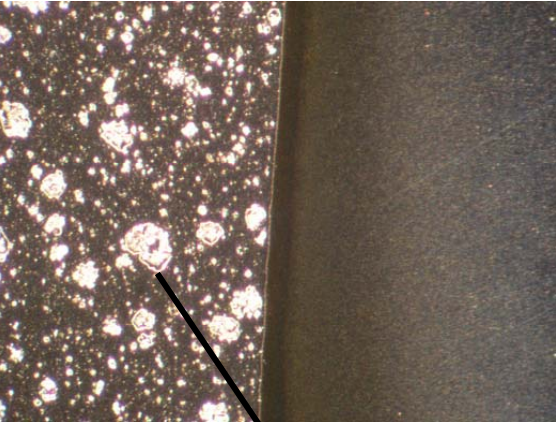


- Red Dots - shows roughening threshold for PM W unheated -  $\sim 1 \text{ J/cm}^2$
- Every other metal  $< 1 \mu\text{m Ra}$  up to 1000 pulses (Re shown)
- Heated PM W Snead at  $1.1 \text{ J/cm}^2$  very smooth except for Divits
- Heated Shwarzkopf PM W @  $2.5 \text{ J/cm}^2$  reaches Snead RT roughness at 1000 pulses

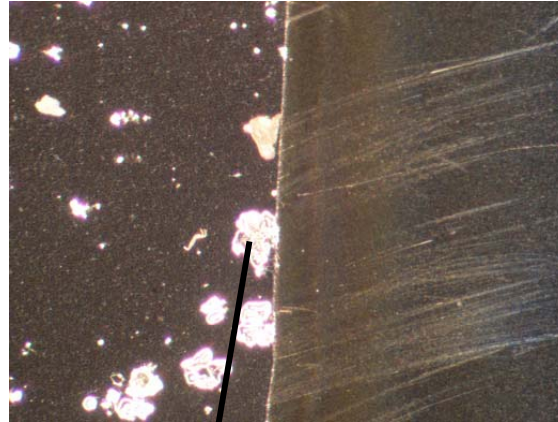
Note: Heated W, W25Re Ra does NOT include Divits

# RHEPP-1 Surface Roughening

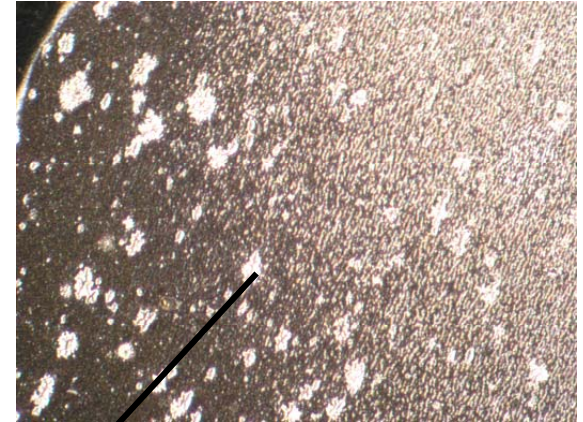
## Current series, Optical Micrographs, surfaces after 400 pulses



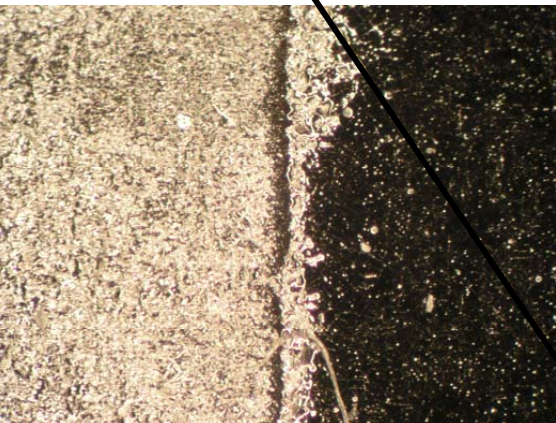
PMW, 1.2 J/cm<sup>2</sup>, 520C



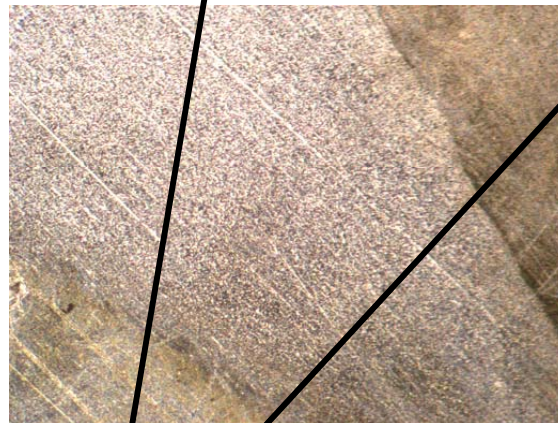
W25Re, 1.2 J/cm<sup>2</sup>, 520C



SingXtal W, 1.2 J/cm<sup>2</sup>, 520C



PMW, 1.2 J/cm<sup>2</sup>, RT



Re, 1.2 J/cm<sup>2</sup>, RT

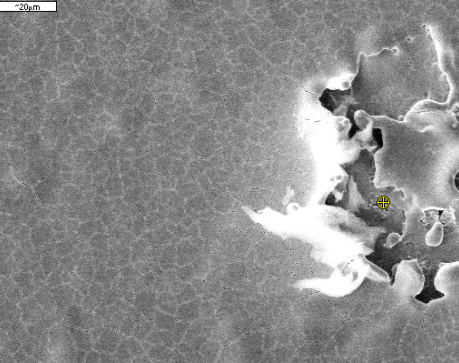


R6650 Graphite, 1.2 J/cm<sup>2</sup>, 520C

What are these?

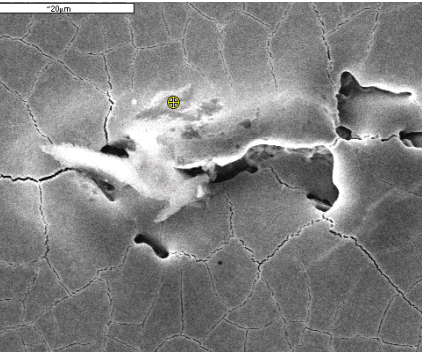
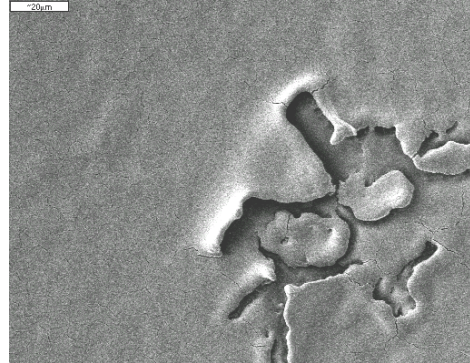
# RHEPP-1 Surface Roughening

## Heated W, W25Re, SingXtal smooth except for 'Divits'

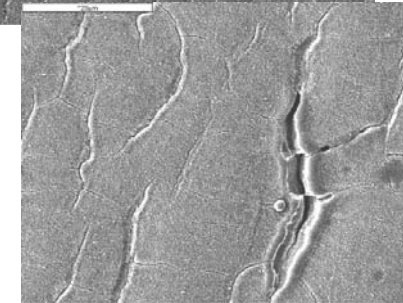
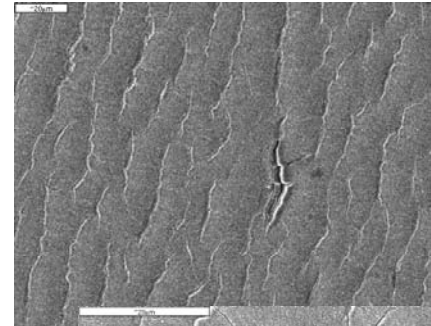


PMW, 1.2 J/cm<sup>2</sup>, 520C, 750Mag

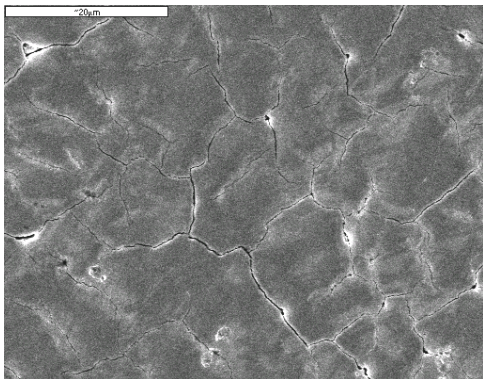
Divit Height ~ 10 μm



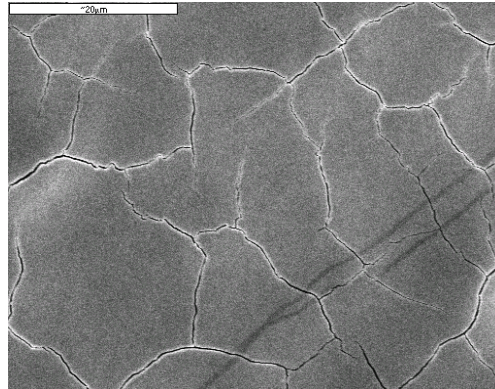
W25Re, 1.2 J/cm<sup>2</sup>, 520C,  
2000Mag



SingXtal W, 1.2 J/cm<sup>2</sup>, 520C,  
750Mag and 2000Mag



(Comparison) Re, 1.2 J/cm<sup>2</sup>, RT



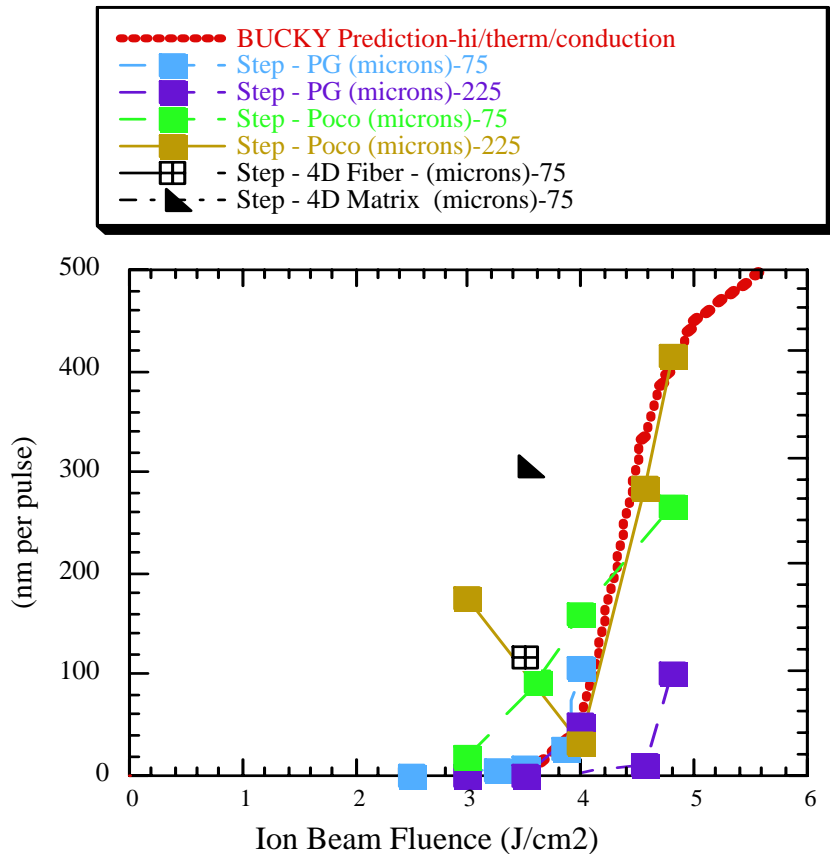
Re, 2 J/cm<sup>2</sup>, RT

Re RT: Microcracks, no Divits

PM W, W25Re, SingXtal W:  
Tears in surface, opening below-grade  
EDS: no contamination

## RHEPP-1 Carbon Response

# Graphite Response: Rapid material removal above sublimation point, consistent with BUCKY predictions



- Mechanically polished pyrolytic graphite (PG), Poco, and 4D carbon composite weave exposed to 75 pulses/225 pulses of 70% C /30% H beam at doses of 1.9 to 5 J/cm<sup>2</sup>
- PG ablation threshold ~ 4 J/cm<sup>2</sup>
- Poco ablation threshold ~ 3 J/cm<sup>2</sup>
- Above threshold, rapid increase in ablated material per pulse with dose. Data scatter reflects uncertainty in dose
- Composite matrix ablates more than PG/Poco, fibers comparable (sample rough)

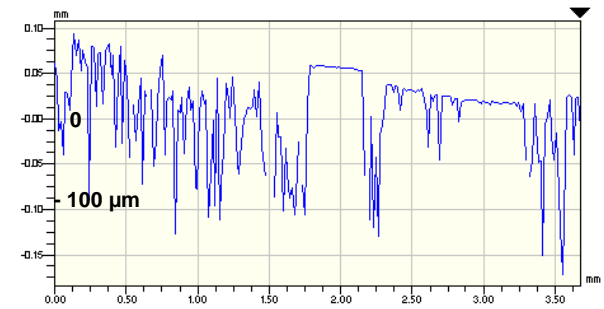
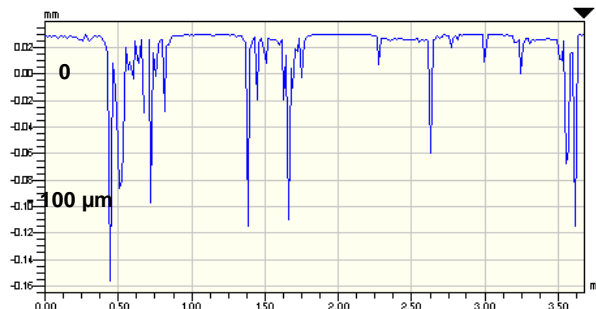
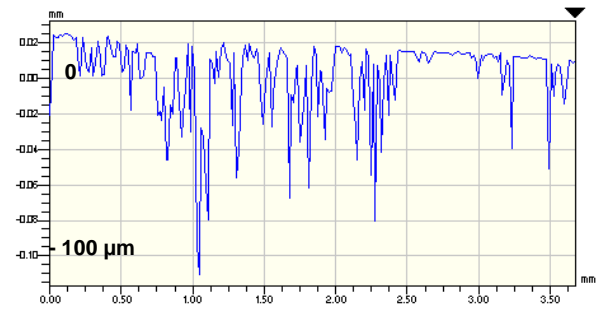
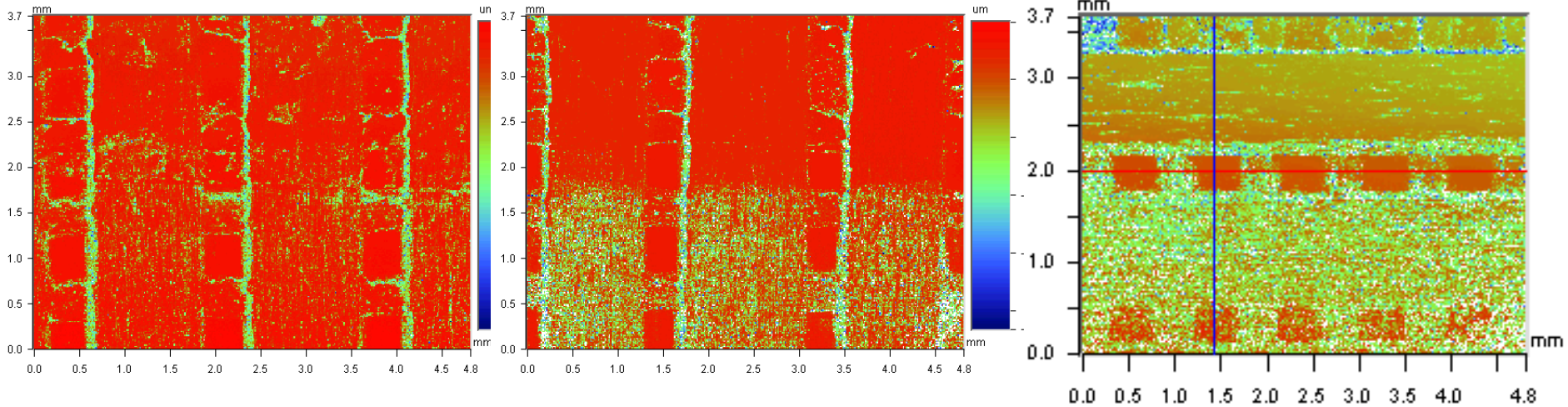
**RHEPP-1 CFC Carbon Response**

**FMI-222 Fiber ends appear ablation-resistant;  
Matrix loss ~ 0.3  $\mu\text{m}/\text{pulse}$  at 4.0  $\text{J}/\text{cm}^2$ :**

Treated at 1.6  $\text{J}/\text{cm}^2$  1000X

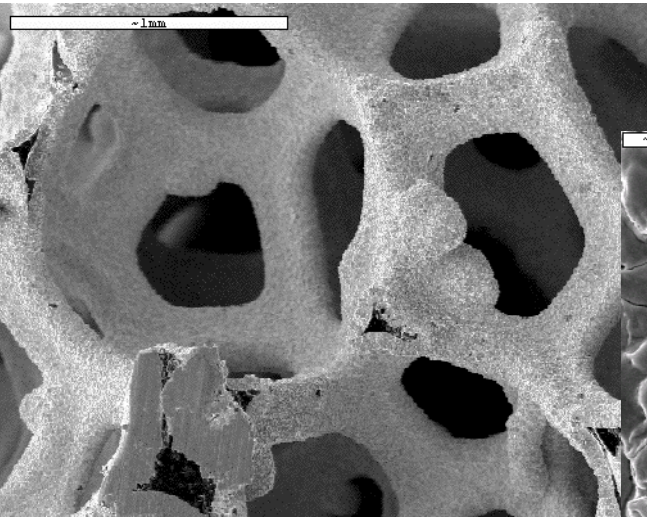
Treated at 2.6  $\text{J}/\text{cm}^2$  600X

Treated at 4.0  $\text{J}/\text{cm}^2$  600X

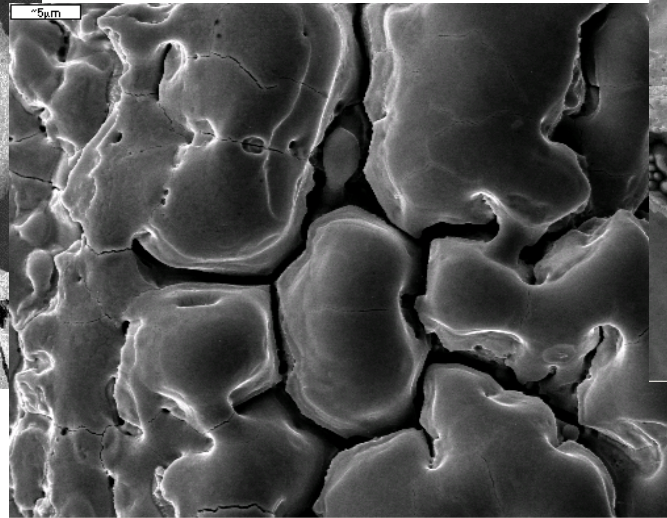


**MIDDLE is approximately  
sublimation threshold**

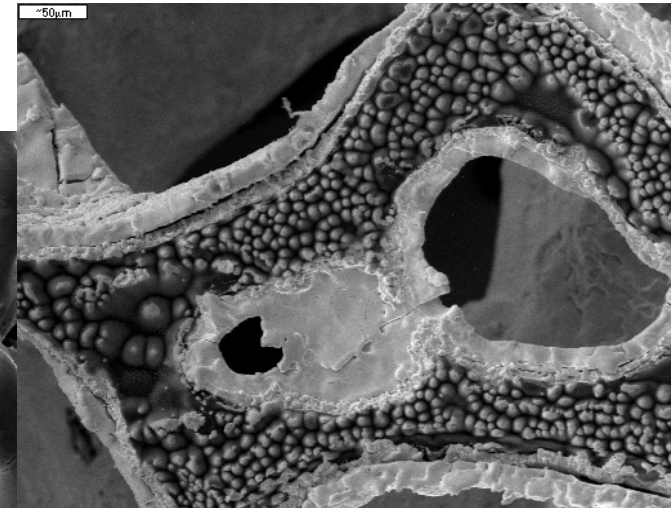
# 'Foam' (Ultramet) exposure to ions: W/TaC suffers erosional loss, W/HfC brittle failure



Foam geometry,  
untreated, 50X

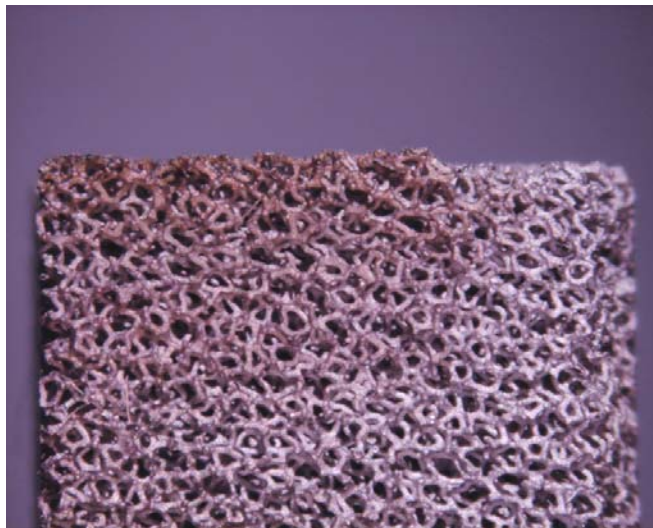


W/TaC, after 1200  
pulses, 2500X



W/HfC, 800  
pulses, 250X

All exposures between  
1 - 1.5 J/cm<sup>2</sup>



Side view, W/TaC,  
after 800 pulses,  
showing  
**EXPANDED**  
height (left)

**SEMs:**  
Thinner structure, better  
Bonding may be necessary

# ESLI CCV/W Test Specimen

## ESLI Carbon-Carbon Velvet (CCV) with sputtered-W Coating

Carbon fiber volume packing 1.9%

Sputtered-W film thickness 1.6  $\mu\text{m}$  over area

Thick (1.6  $\mu\text{m}$ ) caps on fiber tips; thin (<0.1  $\mu\text{m}$ ) film on fiber shafts (thickest at top)

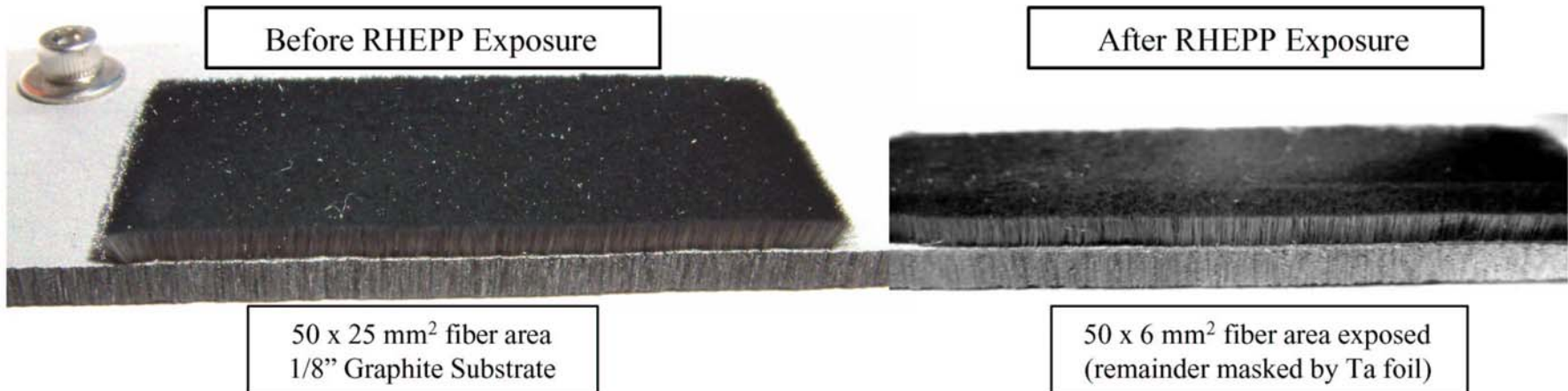
## Design for RHEEP incidence angle (0.3-0.5 radian)

### Fluence distribution

On flat fiber tops 100% of incident fluence

On fiber shafts < 25% of incident fluence (max near top)

On graphite substrate <5% of incident fluence

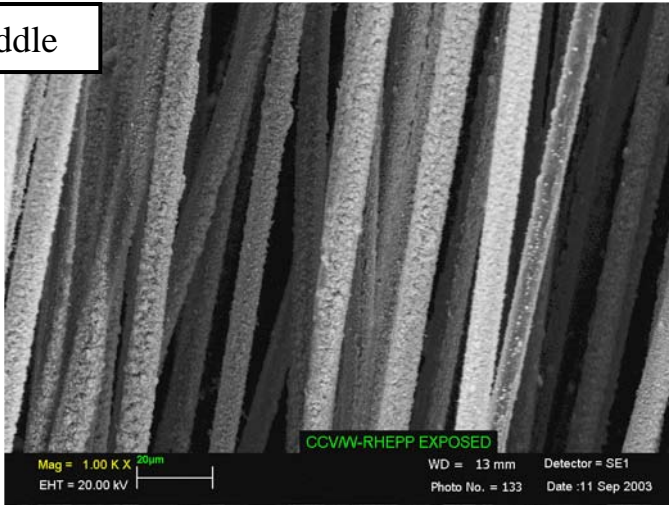




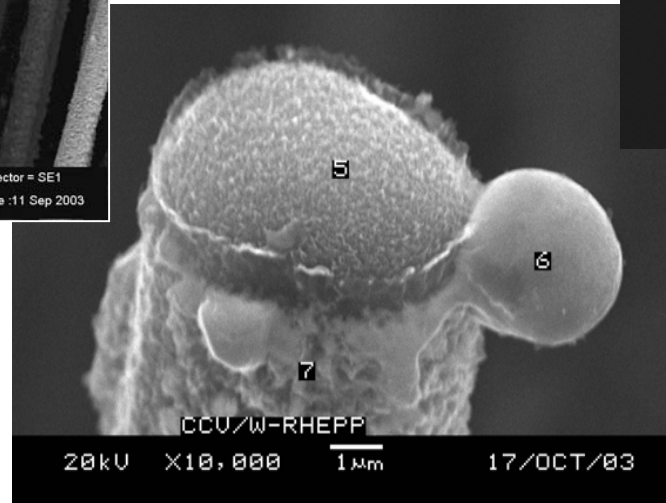
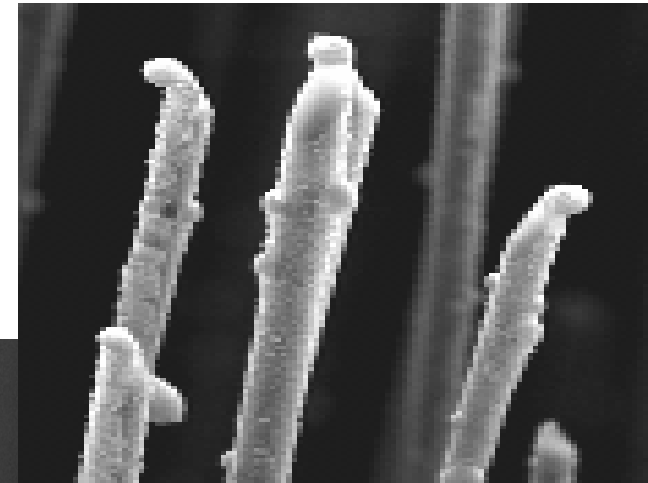
RHEPP-1 Carbon/W Velvet Response

W-coated Carbon 'Velvet' exposure to ions:  
1.6  $\mu\text{m}$  W survives on sharp tips, 200 pulses at 6 J/cm<sup>2</sup>

Middle



Fiber sides, W-sputtered, after treatment



1.6  $\mu\text{m}$  W survives ablation-level  
Fluence on SOME tips

# Summary

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- Ion and X-ray exposure forms key part of Laser IFE Direct Drive First-Wall Threat. RHEPP-1 ion beams (multi-pulse) and Z x-rays (single-pulse) can simulate this threat with good fidelity
- Z results (X-rays on tungsten): melt threshold at  $1.3 \text{ J/cm}^2$ ; roughening threshold is grain-structure dependent, but  $< 1 \text{ J/cm}^2$

## RHEPP Results:

- For W materials, most serious issues are 1) Roughening below melt, evidently caused by thermomechanical stress, and 2) Stress cracking. Stress cracks deepen with pulse number, and may reach deep into interior. This is true regardless of melt. 'Divit' formation may indicate outright mass loss. Lower limit for PM W roughening is low  $\sim 1 \text{ J/cm}^2$ .
- For Graphite/Carbon Composites (CFCs), good survivability at low fluence, but rapid loss above sublimation. Physical sputtering or radiation-induced sublimation?
- Engineered materials such as 'Velvet' or 'Foam' are up-optimized so far, but may be an alternative surface design
- These results are also relevant to MFE W exposure due to Type I EL